

Scalable water splitting on particulate photocatalyst sheets with a solar-to-hydrogen energy conversion efficiency exceeding 1%

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Photocatalytic water splitting using particulate semiconductors is a potentially scalable and economically feasible technology for converting solar energy into hydrogen¹⁻³. Z-scheme systems based on two-step photoexcitation of a hydrogen evolution photocatalyst (HEP) and an oxygen evolution photocatalyst (OEP) are suited to harvesting of sunlight because semiconductors with either water reduction or oxidation activity can be applied to the water splitting reaction^{4,5}. However, it is challenging to achieve efficient transfer of electrons between HEP and OEP particles^{6,7}. Here, we present photocatalyst sheets based on La- and Rh-codoped SrTiO₃ (SrTiO₃:La,Rh; ref. 8) and Mo-doped BiVO₄ (BiVO₄:Mo) powders embedded into a gold (Au) layer. Enhancement of the electron relay by annealing and suppression of undesirable reactions through surface modification allow pure water (pH 6.8) splitting with a solar-to-hydrogen energy conversion efficiency of 1.1% and an apparent quantum yield of over 30% at 419 nm. The photocatalyst sheet design enables efficient and scalable water splitting using particulate semiconductors.

The photocatalyst sheet presented in this work, which consists of SrTiO₃:La,Rh (La/(La+Sr) = Rh/(Rh+Ti) = 4 mol%) as a HEP and BiVO₄:Mo (Mo/V = 0.05 mol%) as an OEP embedded into a Au layer, is hereafter abbreviated as SrTiO₃:La,Rh/Au/BiVO₄:Mo. This sheet was fabricated by the particle transfer method (Fig. 1a and Methods; refs 9,10). In this process, both SrTiO₃:La,Rh and BiVO₄:Mo particles were embedded into a gold layer, as seen in the top-view scanning electron microscopy and energy dispersive X-ray fluorescence spectroscopy (SEM-EDX) mapping images of the SrTiO₃:La,Rh/Au/BiVO₄:Mo photocatalyst sheet (Fig. 1b-f). Photodeposition was then used to modify the photocatalyst sheet with Ru species as a co-catalyst¹¹. Nanoparticulate Ru species were deposited on the surfaces of SrTiO₃:La,Rh and BiVO₄:Mo particles, as shown in Supplementary Fig. 1, and were probably attributable to Ru and RuO_x, respectively, by X-ray photoelectron spectroscopy (XPS) analysis (see Supplementary Fig. 2 and the accompanying discussion) and the Pourbaix diagram of Ru in water system¹².

The mechanism involved in the water splitting reaction on the SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet is illustrated in Fig. 1g (ref. 9).

When SrTiO₃:La,Rh/Au/BiVO₄:Mo is irradiated with visible light, photoexcited electrons are generated in the conduction bands of SrTiO₃:La,Rh and BiVO₄:Mo, whereas positive holes are generated in the donor levels formed by Rh³⁺ ions of SrTiO₃:La,Rh and in the valence band of BiVO₄:Mo. Electron transfer occurs from the conduction band of BiVO₄:Mo to the donor levels of SrTiO₃:La,Rh via gold. Meanwhile, the excited electrons in SrTiO₃:La,Rh reduce water to hydrogen on Ru species serving as a hydrogen evolution co-catalyst, and holes in BiVO₄ oxidize water to oxygen with the aid of RuO_x species functioning as an oxygen evolution co-catalyst, achieving overall water splitting. Therefore, the ability of the photocatalyst sheet for overall water splitting should be promoted by facilitating charge transfer through the underlying Au layer. Note that the Au layer is too thick (approximately 350 nm) to induce the surface plasmon resonance below 520 nm, where the SrTiO₃:La,Rh/Au/BiVO₄:Mo system shows water splitting activity⁹.

To reduce the contact resistance between the semiconductors and the metal layer, the photocatalyst sheet samples were annealed in air^{13,14}. The activity in Z-scheme water splitting was indeed strongly dependent on the annealing temperature and duration, as shown in Fig. 1h and Supplementary Fig. 3; it was maximized (almost doubled) on annealing at 573 K for 20 min. The sample annealed before the deposition of Au (referred to as 'pre-annealed') exhibited similar gas evolution rates to the unannealed sample (Supplementary Fig. 4). This rules out the possibility that the heat treatment intensified the junction between the two kinds of semiconductor particles or altered the semiconducting properties of the photocatalysts favourably. The effect of annealing was further investigated through photoelectrochemical measurements of SrTiO₃:La,Rh/Au and BiVO₄:Mo/Au electrodes. The SrTiO₃:La,Rh/Au and BiVO₄:Mo/Au electrodes generated cathodic and anodic photocurrents, respectively (see Supplementary Fig. 5 and the accompanying discussion). Compared with the pristine electrodes, the electrodes annealed at 573 K for 20 min generated almost twice the photocurrent. These results indicate that the resistance between SrTiO₃:La,Rh and BiVO₄:Mo particles via the Au layer was reduced by annealing at 573 K for 20 min. However,

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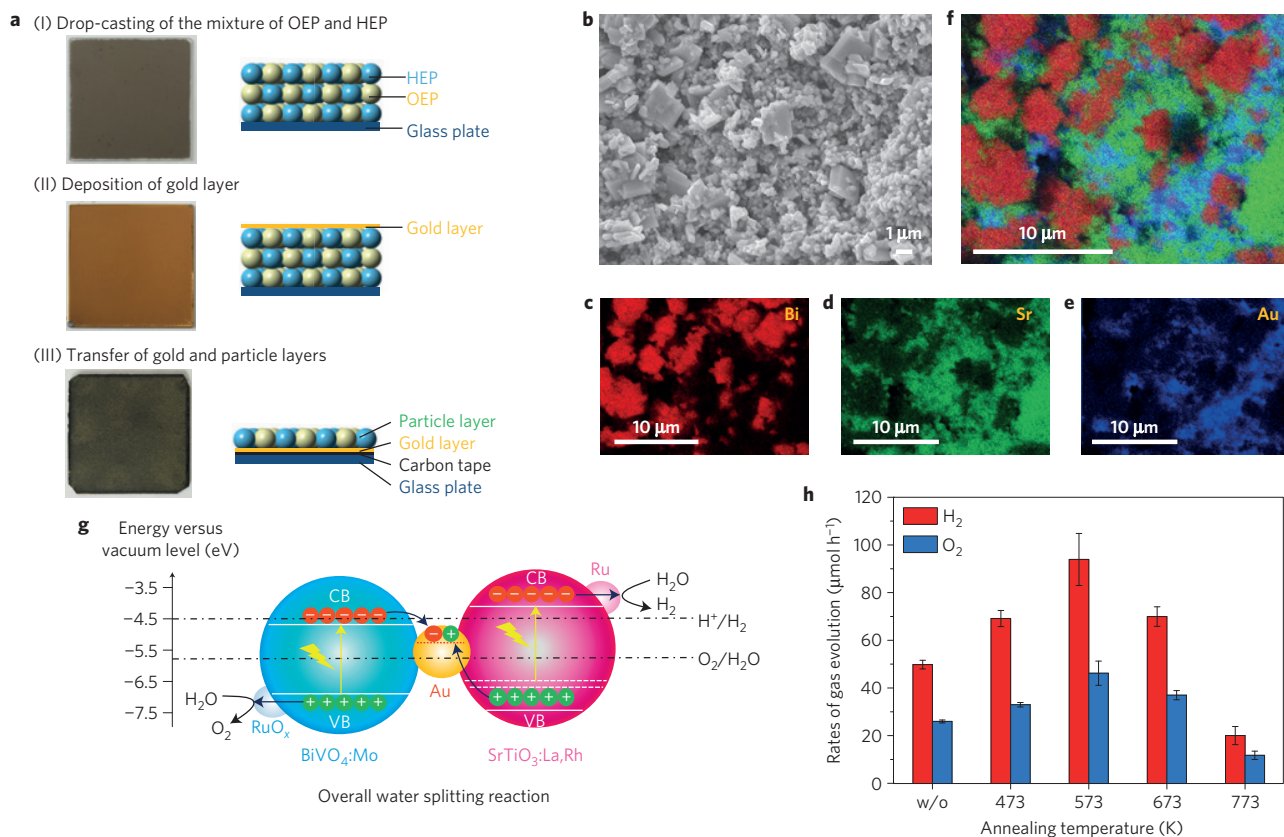


Figure 1 | SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet prepared by particle transfer method. **a**, Illustration of the preparation of the SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet by the particle transfer method. **b–f**, Top-view SEM–EDX elemental mapping images showing an SEM image (**b**), Bi distribution (**c**), Sr distribution (**d**), Au distribution (**e**), and a superimposition of the distributions (**f**). **g**, Schematic of overall water splitting on the Ru-modified SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet. CB, conduction band; VB, valence band. **h**, Dependence of overall water splitting activity of Ru-modified SrTiO₃:La,Rh/Au/BiVO₄:Mo photocatalyst sheets on annealing temperature. All samples were annealed for 20 min. ‘w/o’ stands for the pristine sample. The error bars show the standard deviations. Reactions were carried out under Xe lamp (300 W) illumination ($\lambda > 420$ nm) at 288 K and 5 kPa.

the water splitting activity decreased sharply when the sample was heated at higher temperatures (673–773 K) and/or for longer durations (30 min), probably because of the volatilization of V₂O₅ (refs 15,16). EDX analysis (Supplementary Table 1) revealed that the Bi/V molar ratio of excessively heated samples exceeded that for stoichiometric BiVO₄ (Bi/V = 1). These samples exhibited a lighter yellow colour than the other samples, as illustrated in Supplementary Fig. 6. Notably, the Au layer deposited by electron beam evaporation under heating *in vacuo* tended to produce approximately 20% higher water splitting activity than that deposited by vacuum evaporation without substrate heating. This is probably due to the *in situ* desorption of adsorbents on the photocatalyst particles before Au deposition.

The overall water splitting rate on the Ru-modified SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet (7.5 cm²) annealed at 573 K for 20 min was maximized when the amount of RuCl₃ added was 0.2 μmol (Supplementary Fig. 7), because the Ru species loaded on the photocatalysts not only served as co-catalysts but also blocked incident photons¹¹. The solar-to-hydrogen energy conversion efficiency (STH) of the optimally modified sample was 0.29% at 288 K in the initial three hours of the reaction (Supplementary Fig. 8). However, it was noted that the water splitting reaction competed with the formation of water from hydrogen and oxygen and the photoreduction of oxygen molecules, typical backward reactions, on the Ru-modified SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet. The water splitting rate decreased with irradiation time, although it recovered on degassing the reaction system (Supplementary Fig. 8). In addition, the photocatalytic water splitting activity dropped

to 23% and 10% when the background reaction pressure was increased from 5 to 10 and 20 kPa, respectively (Supplementary Fig. 9), because the hydrogen and oxygen bubbles needed larger numbers of gas molecules to desorb from the photocatalyst sheet at a higher background pressure.

We previously found that Cr₂O₃ shell capping noble metal nanoparticles suppressed the backward reactions whilst maintaining the function of the noble metal as a hydrogen evolution catalyst^{17,18}. Hence, Cr₂O₃ was deposited on the Ru-modified SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet sample by photodeposition. The activity improved with increasing amount of K₂CrO₄ up to 0.1 μmol for 7.5 cm²-sized samples (Supplementary Fig. 10). More importantly, after the photodeposition of Cr₂O₃, the photocatalyst sheet sample maintained a water splitting activity of 78% and 60% at 10 and 20 kPa, respectively (Supplementary Fig. 9). No noticeable decrease in photocatalytic activity was observed for at least 10 h (Fig. 2a). Moreover, an additional surface modification by recently developed amorphous titanium oxide (a-TiO₂) layers^{19,20} suppressed the background pressure effect further. The photocatalyst sheet samples modified with the Cr₂O₃ and a-TiO₂ layers maintained a water splitting activity of 89% and 86% at 10 and 20 kPa, respectively (Supplementary Fig. 9). Suppression of the backward reactions is a critical factor for the activity of photocatalyst sheets because hydrogen and oxygen are generated in close proximity. Hence, the water splitting reaction was carried out at low background pressures at present. However, this problem can be overcome through the development of surface modifications suppressing the backward reactions effectively.

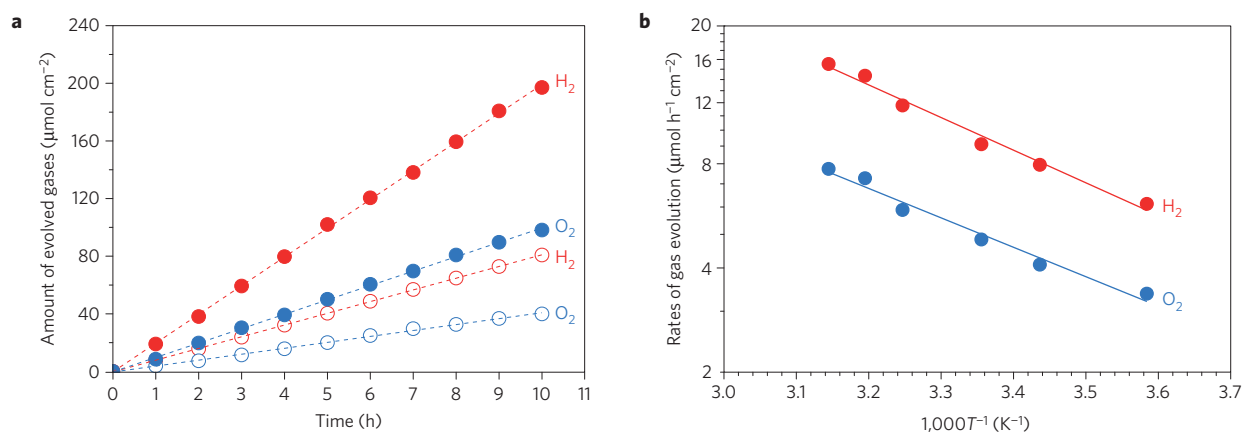


Figure 2 | Effect of reaction temperature on photocatalytic water splitting rate of the SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet. **a**, Time courses of the water splitting reaction on a Cr₂O₃/Ru-modified SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet under simulated sunlight (AM 1.5G) at 288 K and 5 kPa (open symbols) and 331 K and 10 kPa (closed symbols). **b**, Temperature (*T*) dependence of the water splitting activity of the Cr₂O₃/Ru-loaded SrTiO₃:La,Rh/Au/BiVO₄:Mo photocatalyst sheet at a background pressure of 5 kPa under AM 1.5G simulated sunlight. The Au layer was deposited by electron beam evaporation.

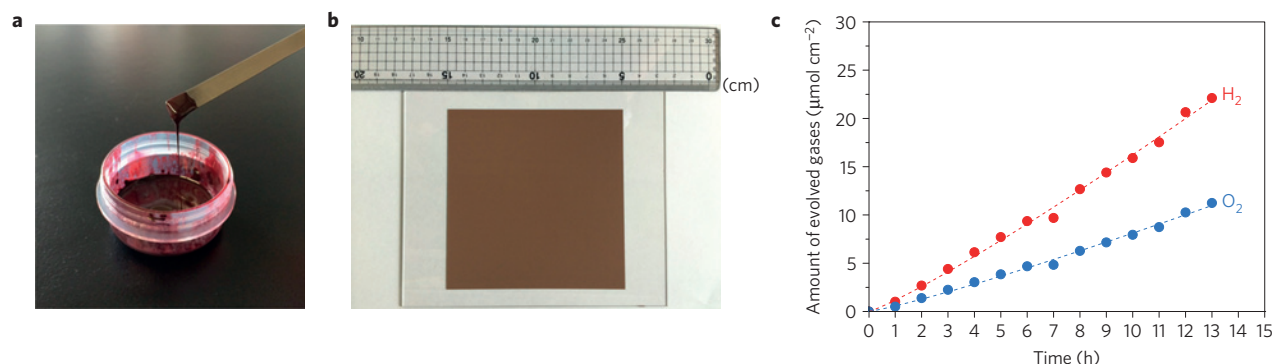


Figure 3 | Printed photocatalyst sheet. **a**, Photograph of the ink used for screen printing the photocatalyst sheet. **b**, Photograph of a 10 × 10 cm SrTiO₃:La,Rh/Au nanoparticle/BiVO₄:Mo printed sheet. **c**, Time course of the water splitting reaction using a Ru-modified SrTiO₃:La,Rh/Au colloidal (40 wt%)/BiVO₄:Mo printed sheet under simulated sunlight at 288 K and 5 kPa. The sample (6.25 cm²) was photodeposited with RuCl₃ · 3H₂O (0.17 μmol).

Figure 2b shows the reaction temperature dependence of the photocatalytic water splitting rate on SrTiO₃:La,Rh/Au/BiVO₄:Mo modified with the co-catalyst Cr₂O₃/Ru. The water splitting reaction was enhanced monotonically with increasing reaction temperature in the range from 279 to 318 K at a background pressure of 5 kPa. The apparent activation energy of the overall water splitting reaction was 18 kJ mol⁻¹, which was higher than that observed for Rh_{2-*y*}Cr_{*y*}O₃-loaded (Ga_{1-*x*}Zn_{*x*})(N_{1-*x*}O_{*x*}) (8 kJ mol⁻¹; refs 21,22). It is plausible that electron transfer across the Schottky barrier at the BiVO₄:Mo/Au interface and hole transfer through the donor levels formed by Rh³⁺ ions in SrTiO₃:La,Rh necessitate thermal excitation for effective conduction. Note that the reaction temperature can reach about 333 K under solar illumination in practical operating conditions^{23–25}. The STH of the pure water (pH 6.8) splitting reaction using a Cr₂O₃/Ru-loaded SrTiO₃:La,Rh/Au/BiVO₄:Mo sheet was 1.0% at 318 K at a background pressure of 5 kPa. Under simulated sunlight (AM 1.5 G) irradiation through Pyrex glass, hydrogen and oxygen bubbles evolved vigorously on the photocatalyst sheet immersed in pure water (pH 6.8) and the gas evolution ceased immediately when the illumination was interrupted, as presented in Supplementary Movie 1 with the original playing speed. The apparent quantum yield (AQY) and STH reached 33% at 419 nm and 1.1%, respectively, at 331 K and 10 kPa (Fig. 2a). By comparison, the photocatalyst suspension Z-scheme system consisting of SrTiO₃:Rh and BiVO₄ powders and Fe^{3+/2+} redox couples exhibited an AQY of 4.2% at 420 nm and an STH of 0.1% under the best reaction

conditions²⁶. Clearly, the photocatalyst sheet recorded by far the highest AQY and STH amongst Z-scheme water splitting systems based on particulate semiconductors, in the absence of any external bias. Higher STHs were reported for photocatalytic water splitting on single materials such as CoO (ref. 27) and CDots/C₃N₄ (ref. 28), although further studies seem to be necessary to confirm the reproducibility.

The present photocatalyst sheet design can be extended to scalable solution processes—for example, screen printing of ink containing an HEP, OEP, and colloidal conductor such as Au and indium tin oxide (Fig. 3a and Methods)—as long as the HEP and OEP particles are bridged in proximity. This is because, unlike in (photo)electrochemical systems, a continuous conductive layer is not needed, owing to the absence of an external electric circuit. Indeed, a 10 × 10 cm printed photocatalyst sheet was readily prepared by the screen-printing method, as shown in Fig. 3b. In a printed SrTiO₃:La,Rh/Au colloidal (10 wt%)/BiVO₄:Mo photocatalyst sheet, Au nanoparticles with sizes ranging from several nanometres to 50 nm were dispersed on the surfaces of the SrTiO₃:La,Rh and BiVO₄:Mo particles, as shown in Supplementary Fig. 11. Au nanoparticles aggregated in significant numbers and bridged the HEP and OEP particles effectively (Supplementary Fig. 12). The printed SrTiO₃:La,Rh/Au colloidal/BiVO₄:Mo sheet produced hydrogen and oxygen from pure water (pH 6.8) in the stoichiometric ratio. The water splitting activity of the printed sheet (Supplementary Fig. 13) was enhanced with increasing amounts

of Au colloid up to 40 wt%, indicating that Au nanoparticle agglomerates effectively served as an electron conductor between $\text{BiVO}_4\text{:Mo}$ and $\text{SrTiO}_3\text{:La,Rh}$. The STH of the Ru-loaded printed $\text{SrTiO}_3\text{:La,Rh/Au}$ colloid (40 wt%)/ $\text{BiVO}_4\text{:Mo}$ sheet was 0.1% on the basis of the average gas evolution rate during the 13-h reaction (Fig. 3c). This value was rather low because of the difficulty in photodepositing co-catalysts on a thick particle layer (approximately $10\ \mu\text{m}$) and the presence of a large amount of Au. Nevertheless, the STH obtained here is still very high compared to those of previously reported particulate photocatalytic systems^{6–8} operating in pure water.

It is notable that the photocatalyst sheet exhibited a high water splitting activity in pure water without any supporting electrolytes, buffering reagents, pH adjustment, or applied voltage. On the photocatalyst sheet sample, HEP and OEP particles are physically embedded into a conductor in the immediate vicinity. This configuration is effective in suppressing the generation of concentration overpotentials of H^+/OH^- and IR drops between the hydrogen and oxygen evolution sites commonly observed in (photo)electrochemical water splitting. Therefore, the photocatalyst sheet with high activity is scalable as is. Another key advance in photocatalyst sheet systems demonstrated by the present study is the facilitation of electron transfer between HEP and OEP by conductor layers or particles. In the $\text{SrTiO}_3\text{:La,Rh/Au/BiVO}_4\text{:Mo}$ photocatalyst sheet system, the annealing-induced enhancement in the electric contact between Au and the semiconductor photocatalysts boosted the gas evolution rates, confirming the importance of electron transfer through the Au layer. In addition, modification of the Ru catalyst with Cr_2O_3 and a-TiO_2 layers effectively suppressed the backward reactions, allowing the photocatalyst sheet systems to maintain a high activity at elevated pressures. The photocatalyst sheet system based on simple and versatile technologies greatly extends the potential of particulate photocatalysts in efficient solar hydrogen production from pure water on a large scale. A gaseous mixture of hydrogen and oxygen is produced on the photocatalyst sheet, but this can be separated safely by introducing the molecular sieving effect of microporous membranes into the system, with a minimum energy loss of 1.1% (refs 29,30). Photocatalyst sheet designs with a relatively low cost could thus lead to advances that would enable industrial solar hydrogen production. The STH of the present $\text{SrTiO}_3\text{:La,Rh/Au/BiVO}_4\text{:Mo}$ system is limited by the short absorption edge wavelengths of $\text{SrTiO}_3\text{:La,Rh}$ and $\text{BiVO}_4\text{:Mo}$ (520 and 540 nm, respectively). Also, the use of Au as a conductor is an economic obstacle to scaling up. The potential of photocatalyst sheet systems can be further extended by employing particulate photocatalysts with narrower bandgap energies and inexpensive conducting materials.

In conclusion, we demonstrated efficient and scalable systems for Z-scheme pure water splitting based on particulate photocatalyst sheets. The $\text{SrTiO}_3\text{:La,Rh/Au/BiVO}_4\text{:Mo}$ photocatalyst sheet exhibited an AQY of 33% at 419 nm and an STH of 1.1% in overall pure water (pH 6.8) splitting. Moreover, a photocatalyst sheet prepared by screen printing an ink containing $\text{SrTiO}_3\text{:La,Rh}$, $\text{BiVO}_4\text{:Mo}$, and a Au colloid exhibited an STH of 0.1%.

Methods

Methods and any associated references are available in the [online version of the paper](#).

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Author contributions

Q.W., T.H., Y.L. and K.D. conceived the photocatalyst sheet design. Q.W. prepared SrTiO₃:La,Rh and the photocatalyst sheet, conducted XRD, DRS, XPS and SEM characterizations and the water splitting reactions. T.H. and K.D. supervised the experimental work. Q.J. prepared the BiVO₄:Mo particles. H.T. prepared the printed photocatalyst sheets. M.Z. and C.W. performed the photoelectrochemical measurements. Q.W., Z.P. and T.T. conducted the surface modification with a-TiO₂. M.N. and N.S. conducted the SEM–EDX elemental mapping measurements. Q.W., Y.L. and I.D.S. carried out the electron beam evaporation. Q.W., T.H., Q.J., H.T., Y.L., A.K., T.Y. and K.D. discussed the results. Q.W. and T.H. wrote the manuscript with contributions from the other co-authors.

Additional information

Supplementary information is available in the [online version of the paper](#). Reprints and permissions information is available online at www.nature.com/reprints. Correspondence and requests for materials should be addressed to K.D.

Competing financial interests

The authors declare no competing financial interests.

Methods

Synthesis of SrTiO₃:La,Rh and BiVO₄:Mo particles. SrTiO₃:La,Rh (La/(La+Sr) = Rh/(Rh+Ti) = 4 mol%) was synthesized by a two-step solid state reaction (SSR; ref. 8). In the initial step, SrTiO₃ was prepared by a conventional SSR. SrCO₃ (Kanto Chemicals, 99.9%, heated in air at 573 K for 1 h before use) and rutile-type TiO₂ (Kanto Chemicals, 99.0%) powders were mixed in a mortar at a Sr/Ti ratio of 1.05 and calcined at 1,423 K for 10 h. Subsequently, the resulting SrTiO₃ was mixed in ethanol with La₂O₃ (Kanto Chemicals, 99.99%, freshly calcined in air at 1,273 K for 12 h before use) and Rh₂O₃ (Kanto Chemicals, 99.9%) and heated at 1,373 K for 6 h.

BiVO₄:Mo (Mo/V = 0.05 mol%) was synthesized through aqueous processes¹⁵. Layered Mo-doped K₃V₅O₁₄ was prepared as a precursor by calcining a mixture of K₂CO₃ (Kanto Chemical, 99.5%), MoO₃ (Kanto Chemical, 99.5%) and V₂O₅ (Wako Pure Chemical, 99.0%) in air at 723 K for 5 h. A suspension of BiONO₃ was prepared by adding Bi(NO₃)₃ · 5H₂O (Kanto Chemical, 99.9%) in distilled water. Mo-doped K₃V₅O₁₄ was added to the suspension of BiONO₃, and the suspension was stirred mildly at 343 K for 10 h. The resulting BiVO₄:Mo (Mo/V = 0.05 mol%) was collected by filtration and washed with distilled water.

The morphologies of the produced samples were characterized by the scanning electron microscope (SEM) images (Supplementary Fig. 1). The phases of the samples were confirmed from the X-ray diffraction (XRD) patterns (Supplementary Fig. 14). The diffuse reflectance spectra (DRS) of the prepared samples are shown in Supplementary Fig. 15.

Preparation of SrTiO₃:La,Rh/Au/BiVO₄:Mo photocatalyst sheets.

SrTiO₃:La,Rh/Au layer/BiVO₄:Mo photocatalyst sheets were fabricated by the particle transfer method^{9,10} by the following procedure: first, a mixture of SrTiO₃:La,Rh and BiVO₄:Mo powders (10 mg each) at the optimized mass ratio of 1:1 (Supplementary Fig. 16) was suspended in isopropanol (99.9%, 0.5 ml) and drop-cast on a glass substrate (3 × 3 cm); then, after drying at room temperature, a thin gold layer (~350 nm) was deposited by thermal vacuum evaporation (VFR-200M/ERH, ULVAC KIKO) at an evaporation rate of approximately 20 nm s⁻¹ under a base pressure of 2.6 × 10⁻³ Pa or electron beam evaporation (Nexdep, Angstrom Engineering) at an evaporation rate of 2 Å s⁻¹ after the chamber was evacuated to a pressure of 2.9 × 10⁻⁴ Pa and the substrate was heated to 673 K; next, the sample was placed on an alumina plate, heated to 473–773 K at 10 K min⁻¹ and kept at the target temperature for 10–30 min; then, the Au film holding the particulate photocatalysts was bonded to a second glass plate (about 3 × 3 cm) with an adhesive carbon tape and then lifted off the primary glass plate; and finally the photocatalyst sheet obtained was ultrasonicated twice, for 2 min each time, in distilled water to remove excess particles piling up on the particle layer, because outer layer photocatalyst particles which did not contact with the Au layer could have little contribution to the photocatalytic activity⁹.

SrTiO₃:La,Rh/Au colloid/BiVO₄:Mo composite prints were fabricated by screen printing. The printing ink was prepared by mixing 60 mg of SrTiO₃:La,Rh powder, 60 mg of BiVO₄:Mo powder, 133–1,200 mg of Au colloid solution (average particle diameter ≈ 20 nm, 10 wt% in ethanol, AUSK-2000E, Shinko Kagaku) and 600 mg of an organic vehicle consisting of α-terpineol, 2-(2-butoxyethoxy)ethanol and acrylic resin ('SPB-TE1', Soken Chemicals and Engineering) at a weight ratio of 62.5:12.5:25.0. After evaporation of ethanol, the paste was screen printed on a glass substrate (about 3 × 3 cm). The printed area was controlled to be 2.5 × 2.5 cm for the measurement of photocatalytic activity. The obtained print was dried at 353 K for 30 min and calcined at 573 K for 30 min.

Preparation of photoelectrodes. Photoelectrodes of SrTiO₃:La,Rh/Au and BiVO₄:Mo/Au were fabricated by a particle transfer method similar to the fabrication of the photocatalyst sheet. SrTiO₃:La,Rh or BiVO₄:Mo powder (20 mg) was dispersed in isopropanol (0.5 ml) by ultrasonication for 30 min. The suspension was deposited on a glass substrate. Au layers 600–800 nm in thickness were deposited on the semiconductor particles by vacuum evaporation repeatedly to maintain the evaporation rate and operation pressure similar to the preparation process of the photocatalyst sheet. The deposited Au film was approximately 3 μm in thickness to ensure the continuity and conductivity. The resulting Au film holding the photocatalyst powder was annealed at 573 K for 20 min, peeled off, and processed into electrodes.

Characterization. The XRD patterns of the samples were measured with a Rigaku RINT-UltimaIII diffractometer using Cu Kα source operated at 40 kV and 40 mA. The DRS were obtained using an ultraviolet–visible–near-infrared spectrometer (V-670, JASCO) and were converted from reflection to absorbance by the Kubelka–Munk method. The SEM images were recorded on the Hitachi SU8020 system at an acceleration voltage of 3 kV. The XPS measurements were carried out using a monochromatic Mg Kα source ($h\nu = 1,253.6$ eV) excited at 8 kV and 10 mA. The analysis chamber pressure was ~10⁻⁶ Pa. The binding energies were corrected using the C 1s peak (285.0 eV) as a reference. A JEOL JSM-7001FA system was used for the SEM–EDX elemental mapping.

Photoelectrochemical measurements. Photoelectrochemical water splitting properties of the SrTiO₃:La,Rh/Au and BiVO₄:Mo/Au photoelectrodes were investigated in the three-electrode configuration using a Ag/AgCl reference electrode (in saturated KCl aqueous solution) and a Pt wire counter electrode. The potential of the working electrode was controlled by a potentiostat (Hokuto Denko, HSV-100). The potential measured to the Ag/AgCl electrode was also expressed in the reversible hydrogen electrode (RHE) scale through calibration. An aqueous solution of 0.1 M Na₂SO₄ was used as an electrolyte solution under stirring and purging with Ar gas. The pH of the electrolyte solution was adjusted to 6.8 by adding H₂SO₄ (aq.) or NaOH (aq.) when necessary. The photoelectrodes were irradiated with a 300 W Xe lamp (Lamp house, R300-3J) equipped with a cutoff filter ($\lambda > 420$ nm). Current–potential curves were measured at a scan rate of 10 mV s⁻¹ from negative to positive potential for the BiVO₄:Mo/Au photoanodes and in the reverse direction for SrTiO₃:La,Rh/Au photocathodes.

Photodeposition of co-catalysts and surface modifiers. The photocatalyst sheets were modified with nanoparticulate Ru species by photodeposition because Ru species was known to work more efficiently as a hydrogen evolution co-catalyst than other metal species in Z-scheme water splitting using SrTiO₃:Rh/BiVO₄ powder suspension¹¹. Subsequently, the photocatalyst sheets were modified with Cr₂O₃ and a-TiO₂ layers successively by photodeposition to suppress the backward reactions. The Cr₂O₃ and a-TiO₂ layers can suppress the backward reactions because oxygen molecules in the reaction solution cannot penetrate the hydrated oxide layers to reach the nanoparticulate noble metal whereas protons and hydrogen molecules can^{17–20}.

In the first step, Ru species were photodeposited on the photocatalyst sheets from RuCl₃ · 3H₂O dissolved in distilled water (40 ml). In the second step, the reactant was changed to an aqueous K₂CrO₄ solution. The amounts of RuCl₃ · 3H₂O and K₂CrO₄ were 0.2 and 0.1 μmol for a photocatalyst sheet sample 7.5 cm² in size unless otherwise noted. In the third step, an a-TiO₂ layer was deposited from a Ti peroxide solution²⁰. The Ti peroxide solution was prepared by the following procedures: titanium tetraisopropoxide (10 μl, Kanto Chemical, 97%) and H₂O₂ (25 μl, Wako Pure Chemical Industries, 30%) were added to distilled water (1 ml). The suspension was subjected to ultrasonication for a few minutes until it turned into a transparent yellow. The photodeposition reactions were conducted in a Pyrex top-irradiation reaction vessel connected to a glass closed gas circulation system. Argon gas was introduced into the circulation system as a carrier gas after removing the air. The total background pressure of argon and water vapour was approximately 5 kPa. The temperature of the reactant solution was maintained at 288 K by cooling water. Photodeposition was performed under a 300 W Xe lamp (Lamp House, R300-3J; irradiation spectra in Supplementary Fig. 17) equipped with a cutoff filter ($\lambda > 420$ nm) until steady gas evolution rates were obtained (typically for 5 h). The number of photons (420 nm < λ < 540 nm) incident on the photocatalyst sheet (7.5 cm²) was measured using a grating spectroradiometer (EKO Instruments Co., Ltd., LS-100) to be 3.9 × 10¹⁷ photon h⁻¹. The amounts of gases produced were measured by gas chromatography (Shimadzu GC-8A with a TCD detector and MS-5A columns and argon carrier gas).

Overall water splitting reaction. Z-scheme water splitting reactions were carried out in the same closed gas circulation system with top illumination from a 300 W Xe lamp equipped with a cutoff filter ($\lambda > 420$ nm) or a solar simulator (Asahi Spectra, HAL-320; irradiation spectra in Supplementary Fig. 18). The photocatalyst sheet samples (about 7.5 cm²) were placed at the bottom of the reaction cell containing pure water (40 ml). The reaction temperature was kept at 288 K and the background pressure was 5 kPa unless otherwise noted. The temperature of the reactant solution was raised (lowered) by a water (ice) bath.

Quantum yield measurement. The apparent quantum yield (AQY) of the Z-scheme water splitting based on a two-step photoexcitation process is given by equation (1):

$$\text{AQY}(\%) = [4 \times n(\text{H}_2)] / n(\text{photons}) \times 100 \quad (1)$$

where $n(\text{H}_2)$ and $n(\text{photons})$ denote the number of produced H₂ molecules and the number of incident photons, respectively. The water splitting reaction was carried out using the same experimental set-up, except for the use of a bandpass filter with central wavelength of 418.6 nm and full-width at half-maximum of 9.5 nm. The spatial distributions of the power spectra are presented in Supplementary Fig. 17. The number of incident photons illuminated on the photocatalyst sheet was measured using a grating spectroradiometer to be 2.9 × 10²⁰ photon h⁻¹ (1.7 mA cm⁻² as the current density if the photon-to-current conversion efficiency was unity).

Solar-to-hydrogen energy conversion efficiency (STH) measurement. The water splitting reaction was carried out under simulated sunlight illumination. The STH is given as

$$\text{STH}(\%) = (R(\text{H}_2) \times \Delta G_r) / (P \times S) \times 100$$

where $R(\text{H}_2)$, ΔG_r , P and S denote the rate of hydrogen evolution during the water splitting reaction, the reaction Gibbs energy of the water splitting reaction ($\text{H}_2\text{O}(\text{l}) \rightarrow \text{H}_2(\text{g}) + 1/2 \text{O}_2(\text{g})$), the energy intensity of the AM1.5G solar irradiation (100 mW cm^{-2}) and the irradiated sample area (7.5 cm^2), respectively. Under a pressure of p , bubbles generated on a photocatalyst sheet contain hydrogen and oxygen at pressures of $(2/3)p$ and $(1/3)p$, respectively, if the water

(2) pressure is ignored. If hydrogen and oxygen behave as ideal gases, ΔG_r is calculated to be 226, 220, and 220 kJ mol^{-1} at 288 K and 5 kPa, 318 K and 5 kPa, and 331 K and 10 kPa, respectively. The light source used closely approximates the power spectrum of the standard AM1.5G illumination as shown in Supplementary Fig. 18. The energy intensities of the simulated sunlight and the standard AM1.5G were 79.3 and 79.1 mW cm^{-2} over the wavelength range 350–1,100 nm, and 23.2 and 25.5 mW cm^{-2} over 350–540 nm, respectively.